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Effect of nickel impurities on charge-density-wave formation in TaS₃

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Introduction of nickel impurities into TaS_3 has the effect of smearing the charge-density-wave transitions in TaS_3 of both orthorhombic and monoclinic phases. The orthorhombic doped crystals show non-Ohmic conductivity beyond threshold fields that are much larger than the threshold field of the pure phase, possibly due to additional pinning of the charge-density wave by nickel impurities. The charge-density-wave state is completely destroyed as 10 at. % or more Ta atoms are replaced by Ni atoms and metallic behavior between 4.2 and 300 K is observed in these cases.

I. INTRODUCTION

The quasi-one-dimensional fibrous compound TaS_3 exists in two phases—orthorhombic¹ and monoclinic.² The orthorhombic phase has a large unit cell containing 24 Ta chains and having the following dimensions: a=36.804 Å, b=15.173 Å, and c=3.340 Å. The space-group symmetry is $C222_1$. This phase shows a commensurate charge-density-wave (CDW) transition at 215 K.³⁻⁵ Recently observations of nonlinear conductivity,^{6,7} frequency-dependent conductivity,⁸ and narrow-band noise⁹ in orthorhombic TaS_3 have been reported. The observed non-Ohmic conductivity has been interpreted in terms of depinning of the CDW's pinned by the lattice, impurities or dislocations at electric fields higher than a threshold field.

Monoclinic TaS₃, on the other hand, is isotypic with NbSe₃ and TaSe₃ (space group $P2_1/m$) with the following unit-cell dimensions:² a=9.515 Å, b=3.3412 Å, c=14.912 Å, and $\beta=109.99^\circ$. This phase exhibits two independent incommensurate CDW transitions at $T_1=240$ K and $T_2=160$ K.¹⁰ Hasegawa *et al.*¹¹ have observed nonlinear conductivity and broad-band noise in monoclinic TaS₃.

Attempts have been made to theoretically understand the effect of impurities on a CDW state.^{12,13} In this paper we report a systematic experimental study of the effect of nickel impurities on the CDW condensate in TaS₃.

II. EXPERIMENTAL METHODS

Crystals of the form $Ni_xTa_{1-x}S_3$ are grown by the vapor-phase transport method from fine powders of constituent elements. The starting material is sealed under vacuum in quartz tubes and heated at 685°C in a furnace for three weeks and then slow-cooled to room temperature in a week. Fibrous crystals of maximum dimensions $10\times0.03\times0.01~\text{mm}^3$ are obtained by this method. There is no apparent distinction in physical appearance between the nickel-doped and the pure TaS₃ crystals.

Stoichiometries of the crystals are determined by the xray fluorescence method. However, this task has been complicated in the case of nickel-doped crystals by the fact that crystals from the same growth tube have somewhat different nickel contents as could be judged from their distinctive resistivity behavior. In addition, the fibrous crystals are extremely thin (\sim 0.01 mm) so fluorescence measurements are not possible on individual crystals. For small Ni concentrations, $x \le 0.08$, it was necessary to determine x in an average way by fluorescence measurements on a number of their crystals. Thus only very approximate x values are known for these crystals. It is estimated that the Ni concentrations of the two "dilute" crystals are given by $x=0.04\pm0.02$, $x=0.08\pm0.02$. The stoichiometry problem is much less severe in the case of Ni_{0.1}Ta_{0.9}S₃ and Ni_{0.13}Ta_{0.87}S₃ which show metallic behavior. Crystals from the same growth tube in these cases show similar resistivity behavior. In these cases we estimate the uncertainty in x to be about ±0.01 .

The dc resistivity is measured along the chain axis by the four-probe method. Silver paint is used to fabricate the current and voltage contacts. The sample is mounted on a "dip probe" which is lowered slowly in a liquid-helium storage Dewar to measure the sample resistance as a function of temperature down to 4.2 K. Dimensions of the sample are measured by a traveling microscope.

III. RESULTS AND DISCUSSION

Figure 1 shows the temperature dependence of the normalized low-field resistivity of pure and nickel-doped TaS_3 samples plotted versus 1000/T in the temperature range 55-300 K. Curve a of Fig. 1 gives the resistivity behavior of pure orthorhombic TaS3. The resistivity increases monotonically with decreasing temperature below the CDW transition temperature of 220 K. The slope of the curve is continually changing with temperature, so that the resistivity does not follow a simple expression of the form $\rho = \rho_0 \exp(\Delta/2k_BT)$. This is consistent with the previously observed resistivity behavior. 6, 10, 14 The roomtemperature resistivity value is $2 \times 10^{-4} \Omega$ cm. The resistivity approaches a saturation value of $\sim 10^2 \Omega$ cm at 4.2 K. We have not observed any monoclinic TaS3 crystal mixed with the pure orthorhombic phase in our growth tube as has been observed by Roucau et al. 10 (their growth temperature is 35°C lower than ours). However, both orthorhombic and monoclinic varieties are present in the same growth tube in the case of nickel-doped crystals.

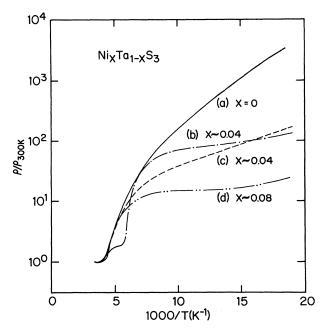


FIG. 1. Resistivity behavior of orthorhombic TaS_3 (a), monoclinic $Ni_{\sim 0.04}Ta_{0.96}S_3$ (b), orthorhombic $Ni_{\sim 0.04}Ta_{0.96}S_3$ (c), and orthorhombic $Ni_{\sim 0.08}Ta_{0.92}S_3$ (d).

Identification of the correct phase, in this case, is done by the temperature dependence of the resistivity. There is only one CDW transition at 220 K in orthorhombic crystals and the resistivity varies smoothly with temperature below this transition temperature. On the other hand, there are two distinct CDW transitions at $T_1 = 240$ K and $T_2 = 160$ K in monoclinic crystals. A sample of the orthorhombic phase can be ascertained by the absence of any resistivity anomaly at 160 K. In Fig. 1 curves c and d show the plots of the normalized resistivity versus 1/Tfor $Ni_{\sim 0.04}Ta_{0.96}S_3$ and $Ni_{\sim 0.08}Ta_{0.92}S_3$, respectively, of the orthorhombic phase. Although it is not obvious from Fig. 1, the CDW transition temperatures of these nickeldoped orthorhombic crystals are slightly higher (10-15°C) than the pure crystal. But a more interesting effect of the nickel impurity is the systematic lowering of the slopes of the curves with increasing nickel content. This would mean that a tendency toward metallic behavior is being induced by the nickel impurities. This suggests that the long-range coherence of the CDW state is being partially destroyed by the impurities. A similar effect is also observed in monoclinic Ni ~0.04Ta_{0.96}S₃ as shown in curve b of Fig. 1. In this case, however, there is no noticeable change from the pure crystal in the two transition temperatures. Only the slope of the curve is lower than in the pure monoclinic phase. 10,11 With further increase of nickel impurities, the coherence of the CDW state is completely destroyed and the doped crystals ultimately become metallic. Figure 2 shows this metallic behavior for Ni_{0.1}Ta_{0.9}S₃. Qualitatively similar behavior is observed for Ni_{0.13}Ta_{0.87}S₃. Identification of the actual phases of individual metallic crystals is not possible from the resistivity behavior alone since the resistivity anomalies of the CDW state are now absent. Both phases, orthorhombic and monoclinic, a mixture of the two, or a

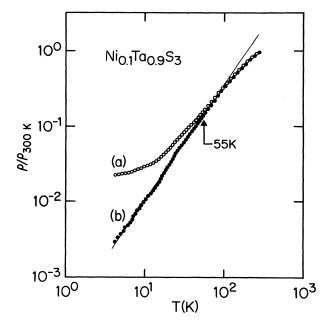


FIG. 2. Temperature dependence of the resistivity of $Ni_{0.1}Ta_{0.9}S_3$ as measured (a) and with the residual resistivity subtracted (b).

new phase may be present among these metallic crystals. Given the complex structures of pure TaS₃ with many atoms per unit cell and the fact that we were able to obtain only very small quantities of these crystals, it was not possible to determine the unit-cell parameters of the metallic crystals by x-ray measurements. The absolute resistivity values of these crystals at room temperature are comparable to those of pure TaS₃ within our accuracy of measurements which is, of course, not very good due to difficulties in measuring the dimensions of these thin crystals. The resistivity ratio, $\rho_{300}/\rho_{4.2 \text{ K}}$, is ~50 for $Ni_{0.1}Ta_{0.9}S_3$ and ~20 for $Ni_{0.13}Ta_{0.87}S_3$. These relatively high ratios suggest that there may well be considerable order in the Ni atomic positions; that is, the Ni atoms may not enter the lattice in a random fashion. The curve a of Fig. 2 shows the temperature dependence of the normalized resistivity and the curve b is obtained after subtracting the residual resistivity which has been estimated by extrapolating the curve a to T=0 K. The linear behavior of the curve b in the low-temperature region suggests that the resistivity of the metallic crystals can be expressed as $\rho(T) = \rho_0 + CT^n$, where ρ_0 is the temperature-independent residual resistivity. The exponent has a value $n=1.5\pm0.1$ for Ni_{0.1}Ta_{0.9}S₃ below 55 K and a value $n=1.3\pm0.1$ for Ni_{0.13}Ta_{0.87}S₃ below 90 K. In both cases the resistivity behavior goes approximately as $T^{0.8}$ at room temperature and the tendency to saturation would likely be seen at a higher temperature. Such saturation effects are not unexpected in metals with resistivity values above about 200 $\mu\Omega$ m.¹⁵

The normalized dc conductivity, measured as a function of dc electric field across the length of the sample, is shown in Fig. 3 for orthorhombic TaS_3 at T=130 K. The conductivity is Ohmic for electric fields less than ~ 1.5 V/cm. Above this threshold field, the normalized

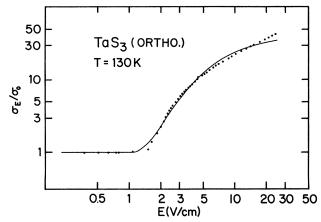


FIG. 3. Electric field dependence of the conductivity of orthorhombic TaS₃ at 130 K. The curve is a fit to Eq. (1) with the parameters given in the text.

conductivity rises steadily up to a factor of 60 for an electric field of 35 V/cm. At electric fields higher than 35 V/cm, there is a steady drift in the value of the sample resistance, indicating sample heating. Our observed field dependence of conductivity of TaS_3 is consistent with the observation of nonlinear conductivity by Thompson, Zettle, and Grüner. These authors have compared their nonlinear conductivity results to the Zener-tunneling model of Bardeen and also to the classical model of Grüner et al. The classical model is inadequate because it predicts that the differential conductivity (dI/dV) diverges at the threshold field, a result not seen experimentally. It was found that the data could be fitted reasonably well by the Bardeen expression for quantum-mechanical tunneling,

$$\sigma(E) = A (1 - E_T/E) \exp(-E_0/E)$$
, (1)

where E_T and E_0 are threshold and single-particle Zener fields, respectively, and A is a constant. Figure 3 also shows a fit to Eq. (1) which was obtained with $E_T = 1.1$ V/cm, $E_0 = 5.5$ V/cm, and A = 44. These parameters are similar to those obtained by Zettl et al.⁶ However, it is seen that there are some quantitative discrepancies between Eq. (1) and experiment, particularly at high fields where sample heating may not be negligible. Of most interest for the present work is the behavior of the non-Ohmic conductivity in the Ni-doped samples. Figures 4 and 5 show this dependence for both orthorhombic and monoclinic Ni_{~0.04}Ta_{0.96}S₃ at 130 K. Also shown are the best fits we could obtain to Eq. (1). The parameters for the orthorhombic and monoclinic crystals are $E_T = 15$ V/cm, $E_0=75$ V/cm, and A=20, and $E_T=1.3$ V/cm, $E_0=6.5$ V/cm, and A=13, respectively. For the orthorhombic case the fit is particularly poor, although it is clear that the threshold fit has gone up by a large factor (~ 14) . The sharp threshold field appears to be lost in this case, thus suggesting inhomogeneities in the electric field caused by the impurities and perhaps a loss of coherence of the CDW. On the other hand, the monoclinic structure with the same nominal Ni concentration shows a threshold field much smaller than that of the orthorhombic phase. These results suggest that the specific struc-

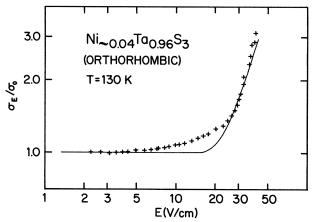


FIG. 4. Electric field dependence of the conductivity of orthorhombic $Ni_{\sim 0.04}Ta_{0.96}S_3$ at 130 K. The curve is a fit to Eq. (1) with the parameters given in the text.

ture, whether the impurity atoms are substitutional or interstitial, plays an important role in the pinning of the CDW. A related point is that Hsieh et al. 18 have found that doping orthorhombic TaS₃ with Nb and Se depresses and broadens the Peierls transition. Of particular note is that these workers found at 130 K $E_T \simeq 0.2$ V/cm for pure TaS₃ (compared to our 1.1 V/cm) and that very small concentrations of isoelectronic impurities such as Nb or Se had a strong effect on E_T . For example, they found that replacing only 2 parts in 1000 of Ta by Nb changed E_T from 0.2 to 15 V/cm. Thus, in comparison, the effect of impurities in our Ni-doped orthorhombic crystals seems relatively weak. One might speculate from this that the Ni impurities, which are not isoelectronic, do not go into the TaS₃ structure as substitutions for Ta. This points toward a need for detailed structural information on the doped crystals, and also for an accurate method of measuring the impurity content of individual crystals in order to derive a quantitative relation between threshold field and impurity concentration. Such information is not easily obtainable for these crystals because of the small size of the doped crystals that can be grown.

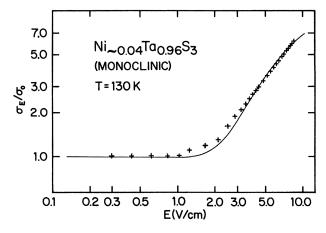


FIG. 5. Electric field dependence of the conductivity of monoclinic $Ni_{\sim 0.04}Ta_{0.96}S_3$ at 130 K. The curve is a fit to Eq. (1) with the parameters given in the text.

In summary, we have shown qualitatively that Ni impurities have the effect of increasing the threshold field for the depinning of CDW's. Rough agreement between the $\sigma(E)$ data and the Bardeen tunneling theory was obtained for nominally pure orthorhombic TaS₃ and monoclinic Ni-doped TaS₃. However, neither the classical model nor the tunneling model is capable of accounting for all of the experiments on TaS₃ and NbSe₃. Recently Tua and Zawadowski have reviewed other theoretical models and introduced a modified classical model to

describe the nonlinear conductivity of CDW systems with strong impurity pinning.¹⁹ Their model predicts that the differential conductivity is nonsingular at the threshold field in agreement with the results presented here and in Refs. 6 and 18.

ACKNOWLEDGMENT

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