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S.Y. Zhang University of Nebraska - Lincoln

Z.S. Shan University of Nebraska - Lincoln

Yi Liu University of Nebraska-Lincoln, yliu@unl.edu

David J. Sellmyer University of Nebraska-Lincoln, dsellmyer@unl.edu

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High Coercivity SmFeSiC Films Fabricated by Multilayer Sputtering

S.Y. Zhang⁽¹⁾, Z.S. Shan⁽¹⁾, Y. Liu⁽²⁾, D.J. Sellmyer⁽¹⁾ Center for Materials Research and Analysis University of Nebraska, Lincoln, Nebraska 68588-0113 T.Y. Zhao, J.G. Zhao, B.G. Shen Institute of Physics, Chinese Academy of Sciences, Beijing, 100080

Abstract-SmFeSiC thin films have been obtained by sputtering SmFe/C(Si) multilayers with a Ta underlayer on Si substrates and subsequently annealing at 700° C. The coercivity of the SmFeSiC films strongly depends on the sputtering and annealing conditions. The influence of the thickness of the Ta underlayer, the thickness ratio of SmFe to C(Si), the argon pressure and heat treatment In- plane coercivities up to 7.2 kOe and squareness of 0.94 were obtained.

I. INTRODUCTION

Permanent-magnet films such as CoCrX (X = Ta, Pt, Ni...) [1], SmCo [2], NdFeB [3], Sm(Fe,T) (T = Ti,V) [4], and SmFeN [5] have been attracting much attention for applications as magnetic recording media, bias magnets and exchange-spring materials. The intermetallic compounds $Sm_2Fe_{17}C_v$ have intrinsic magnetic properties similar to Sm₂Fe₁₇N_x. However, until now studies have been concentrated on bulk $Sm_2Fe_{17}C_y$ materials. No work on SmFeC films has been reported. The thin-film structure provides an additional synthetic route, a better uniformity and certain device applications. We found SmFeC films with high coercivity are difficult to synthesize from only three elements. It was reported that some fourth elements, such as Si,Ga and Al, that partially replace Fe, stabilize the structure of 2:17-type carbides [6,7]. In this paper we report the successful preparation of SmFeSiC films with coercivity as high as 7.2 kOe, and systematic studies of the effects of the process variables on the film properties.

II. EXPERIMENTS

SmFeSiC films were prepared by dc magnetron sputtering. First, SmFe/C(Si) multilayer films were deposited at room temperature on Si substrates with Ta underlayer and overcoating, then annealed in vacuum of 2×10^6 Torr. A composite Sm₂Fe₁₇(Sm) alloy target and a composite graphite(Si) target were used. The chamber base pressure was about 4×10^7 Torr. The deposition rates

(1) Behlen Laboratory of Physics

(2) Department of Mechanical Engineering.

were about $9\dot{A}$ /sec for Sm_2Fe_{17} and about $2\dot{A}$ /sec for C(Si). The sputtering pressures ranged from 5 to 30mT of argon gas.

The magnetic properties of the films were measured by an alternating gradient force magnetometer with a maximum field of 14 kOe. Compositions were measured by a EDX spectrometer, and the structural properties were examined by X-ray diffraction (CuK α), and transmission electron microscopy (TEM).

III. RESULTS AND DISCUSSION

The intrinsic coercivity of SmFeSiC thin films was studied by systematically varying the sputtering and heat treatment parameters in order to find an optimal set of variables that yields a maximum in coercivity. Figure 1 shows that for a given thickness of single SmFe(37Å) layer, and a given argon pressure (10 mT), the coercivity for the films annealed at 700°C for 5 min increases with increasing single C(Si) layer thickness, reaches the maximum value at 10Å, and then decreases with increasing single C(Si) layer thickness. However, nearly the same coercivities are obtained for the samples with a given



Fig. 1 Coercivity vs. thickness of single C (Si) layer for Si/Ta(1000Å)/ [SmFe(37Å)/C(Si)(XÅ)] x58 /Ta(200Å), where "x 58" denotes the no. of bilayers and this notation is used hereafter, films after annealing at 700°C for 5 min.

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Fig. 2 Coercivity vs. argon pressure for Si/Ta(1000Å)/[SmFe(100Å)/C(Si)(27Å)]x21/Ta(200Å)films after annealing at 700°C for 5 min.

ratio of SmFe/C(Si)=3.7 prepared at 5-15mT argon pressure (Fig. 2), above which the coercivity decreases



Fig. 3 Coercivity vs. annealing temperature and time for Si/Ta (1000Å)/[SmFe/C(Si)]/Ta(200Å) films.



Fig. 4 Coercivity and magnetization vs. Ta underlayer thickness for $Si/Ta/[SmFe(100 \text{\AA})/C(Si)](27 \text{\AA})]x21/Ta(200 \text{\AA})$ films after annealing at 700°C for 5 min.

rapidly with increasing the argon pressure. It has been shown that the argon pressure can affect the microstructure of the similar films strongly[8].

Figure 3 illustrates coercivity as a function of annealing temperature and time. It is seen that annealing temperature has important effects on the coercivity. The coercivity for samples #1 and #2, which have the same thickness ratio of SmFe/C(Si)=3.7 and same total multilayer thickness but different single layer thickness, reaches the same maximum value at 700°C(Fig.3a). For the sample #3 which has the thickness ratio of Sm/Fe/C(Si)=2.9, the coercivity reaches the maximum value at 750°C (Fig.3a). The annealing time does not affect the coercivity greatly (Fig. 3b). These optimal values of the ratio of SmFe/C(Si)=3.7, 10mT argon pressure, and annealing at 700°C for 5min for SmFeSiC on Si/Ta(1000Å), were used for further studies.

Figure 4 shows the dependence of coercivity and saturation magnetization for SmFeSiC films on Ta underlayer thickness. The coercivity and magnetization increase very rapidly with increasing thickness of Ta up to 400Å and 200Å respectively, and then remain nearly constant. These behaviors are similar to other alloy systems[9]. It was reported that the grain size and texture of the magnetic layer is controlled by the presence of Cr underlayer[8,9,10]. Consequently the magnetic properties of these films is improved as shown for CoCrPt[10] and SmCo[8]. X-ray diffraction patterns show that the Ta underlayer on Si substrates has a <002> texture after deposition at room temperature.

The composition of optimal films detected by EDX is $Sm_2Fe_{12}Si_2C_x$. X-ray diffraction studies of SmFeSiC layers have been performed and no characteristic crystalline peaks were observed. This may be attributed to the fact that the crystallite size is rather small and the films are fairly thin; thus the films are in the so-called x-ray

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amorphous states. High resolution TEM image shows the grain size is in the range of 10-20 nm and no amorphous



Fig. 5. High resolution TEM image for Si/Ta(1000Å)/[SmFe(37Å)/C(Si)(10Å)x58/Ta (200Å) films after annealing at 700°C for 5 min.

phase was detected (Fig. 5). Several possible phases around the Fe(Si)/Sm atomic ratio of 7 are Cu₇Tb and Zn₁₇Th₂, which are related to each other by structure transformation[10]. However, the electron diffraction ring pattern does not show an acceptable match to the Cu₇Tb structure or the Zn₂Th₁₇ structure. The analysis of the crystal structure of the magnetic phase is in progress by convergent beam electron diffraction and HREM.

The typical hysteresis loop of SmFeSiC films is shown in Fig. 6. SmFeSiC film has an in-plane coercivity of 7.2 kOe, however the SmFe film with same thickness of magnetic layer as SmFeSiC made by same condition has a coercivity of about 2 kOe. The squareness of in-plane loops for SmFeSiC is 0.94, which is better than those of SmFeN films[5]. The hysteresis loops for both in-plane and perpen-



Fig. 6 Hysteresis loop of the sample shown in Fig. 5.

dicular loops are similar, which indicates that SmFeSiC film made by this method is isotropic.

The results reported here show that high coercivity up to 7.2 kOe can be achieved in SmFeSiC thin films. Studies on the role of the Ta underlayer, and the correlation of the coercivity with the microstructure in SmFeSiC films are in progress.

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