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A non-vacuum process for preparing nanocrystalline 

\( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) materials involving an open-air solvothermal reaction

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Abstract

A non-vacuum, two-step process has been used to prepare a series of nanocrystalline \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) \((x = 0, 0.25, 0.5, 0.75, 1)\) materials. An open-air solvothermal preparation in triethyleneetramine solvent was followed by annealing at 500 °C in a nitrogen atmosphere for 20 min. All materials have mixed clustered plate, spherical particle, and nanorod morphologies with the smallest particle diameters ranging between 20 and 40 nm. Raman spectroscopy and X-ray diffraction (XRD) confirm that indium/gallium ratio control is possible over a wide range. The solvothermal reaction step yields a mixture of chalcopyrite and \( \text{Cu}_{2-2x}\text{Se} \). This is converted to pure chalcopyrite product by annealing at 500 °C.

Keywords: chalcopyrites, nanocrystalline, CIGS, solvothermal, processing

1. Introduction

For some time, the chalcopyrite semiconductors \( \text{CuInSe}_2 \) and \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) have been leading thin-film material candidates for incorporation in high-efficiency photovoltaic devices [1–4]. Interest in the development of more cost-effective, non-vacuum film production techniques has stimulated research in the solution-based preparation of nanocrystalline \( \text{CuInSe}_2 \) and \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \). Many reported solvothermal preparations involve the reaction of constituent elements or their salts in heated solution for several hours or days [5–12]. The solvent ethylenediamine \( (\text{C}_2\text{H}_4\text{N}_2) \) has often been employed in the preparation of binary selenides [5–7], \( \text{CuInSe}_2 \) [8–11], and \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) [10, 12]. A strongly coordinating solvent, \( \text{C}_2\text{H}_8\text{N}_2 \) has been proposed to solubilize reactant materials through the formation of solvent complexes such as \( [\text{Cu} (\text{C}_2\text{H}_8\text{N}_2)_2]^{+} \) [11, 12] and \( [\text{Se} (\text{C}_2\text{H}_8\text{N}_2)_3]^{-} \) [13]. Reported \( \text{CuInSe}_2 \) nanorods morphologies have been attributed to the square-planar geometry of \( [\text{Cu} (\text{C}_2\text{H}_8\text{N}_2)_2]^{+} \) serving as a template for one-dimensional growth [11].

While solvothermal reactions in \( \text{C}_2\text{H}_8\text{N}_2 \) may be conducted in air, required temperature conditions vary depending on the complexity of the material prepared. Binary selenides may be prepared without solvent heating [5, 6], however, chalcopyrite crystalline structure formation requires elevated temperatures. \( \text{CuInSe}_2 \), nanorods have been prepared from refluxing \( \text{C}_2\text{H}_8\text{N}_2 \) (120 °C) [11] and several \( \text{CuInSe}_2 \) and \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) nanocrystalline samples have been prepared by superheating reaction mixtures in sealed containers at temperatures ranging from 140 to 280 °C [8–10, 12]. Reaction temperatures in excess of 230 °C were required to form \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) nanoparticles of diameters less than 100 nm [10, 12]. More recently, monodisperse \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) nanoparticles of 15 nm in diameter have been prepared from \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) nanoparticles in pyridine under a nitrogen atmosphere at 0 °C [14].

A two-step process consisting of a solvothermal reaction followed by heat treatment of the isolated solid product has been reported in two cases. Carmalt et al. [15] solvothermally prepared \( \text{CuInSe}_2 \) from \( \text{CuBr} \), \( \text{InCl}_3 \), and \( \text{Na}_2\text{Se} \) in toluene, resulting in amorphous \( \text{CuInSe}_2 \) that converted to crystalline form after annealing at 500 °C for 24 h. Li et al. [5] reported unknown “molecular precursors” following room-temperature reactions of several metals or their salts with \( \text{Se} \) in \( \text{C}_2\text{H}_8\text{N}_2 \) solvent. Heat treatments at 250 °C resulted in crystalline \( \text{Ag}_2\text{Se} \), \( \text{CuSe} \), \( \text{PbSe} \), \( \text{SnSe} \), \( \text{MnSe} \), \( \text{Bi}_2\text{Se}_3 \), and \( \text{Sb}_2\text{Se}_3 \).

Given that the low boiling point of \( \text{C}_2\text{H}_8\text{N}_2 \) is a drawback in potential solvothermal preparations of quaternary chalcopyrites, the use of a chemically similar solvent with a higher boiling point may show greater promise. In this paper, we report the preparation of a series of \( \text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 \) \((x = 0, 0.25, 0.5, 0.75, 1)\) nanocrystalline materials of varying indium–gallium ratios via a two-step process that features an open-air solvothermal reaction in refluxing triethyleneetramine \( (\text{C}_2\text{H}_8\text{N}_2) \) followed by annealing of the resulting solid-state product. With a molecular structure and coordinating ability similar to \( \text{C}_2\text{H}_8\text{N}_2 \) \( \text{C}_2\text{H}_8\text{N}_2 \) greatly...
increases the utility of chelating amine solvent use in this prepa-
ration as well as its scaleup potential. Without the need of an in-
ert atmosphere, expensive commercial selenium source, or ne-
cessity to exceed the normal boiling point of the solvent (267 °C),
this reaction followed by annealing in a nitrogen atmosphere at
500 °C yields CuIn$_{1-x}$Ga$_x$Se$_2$ nanocrystalline materials of vary-
ing indium–gallium ratios. The reaction system shows potential
for generating a variety of ternary and quaternary chalcopyrite
materials in the CuInSe$_2$ family. To the best of our knowledge,
we are the first to report solvothermally prepared CuIn$_{1-x}$Ga$_x$Se$_2$
($x>0$) via open-air means.

2. Experimental

CuIn$_{1-x}$Ga$_x$Se$_2$ product samples are labeled as follows:
CIS ($x = 0$), CIGS-1 ($x = 0.25$), CIGS-2 ($x = 0.5$), CIGS-3 ($x
= 0.75$), and CGS ($x = 1$). Desired stoichiometric quantities of Se
(1.00 mmol), CuCl$_2$ (0.50 mmol), InCl$_3$ (0–1.00 mmol), and GaCl$_3$
(0–1.00 mmol) were refluxed in C$_2$H$_4$N$_2$ at 267 °C for 24 h (CIS)
or 48 h (CIGS-1, CIGS-2, CIGS-3, and CGS). After isolation by
centrifugation, rough films of the products were cast on boro-
silicate glass substrates from methanol and acetone suspensions.
The samples were placed in an enclosed graphite chamber [16].
This was inserted into a quartz tube surrounded by an 8000 W
Quad Ellipse Chamber Heater that was connected to a Model
915 power supply/temperature controller from Research, Inc.
The tube was evacuated, filled with nitrogen (99.99% purity),
and heated to a set point that was varied between 200 and 500 °C
in different experiments with a ramp rate of 8 °C/s. After an-
nealing for 20 min at the set point, the sample was cooled under
flowing N$_2$. The product materials were characterized by micro-
Raman spectroscopy (Horiba/Jon Yvon LabRAM HR8000), AES
(Physical Electronics 560 AES/XPS), XRD (Bruker-AXS D8 Dis-
cover), and SEM (Hitachi S4700).

3. Results and discussion

Upon refluxing the starting materials in C$_2$H$_4$N$_2$, a fine black
precipitate forms within 5 min. This has been isolated and iden-
tified by micro-Raman spectroscopy and XRD as Cu$_2$Se [17].
Over 24–48 h of continued refluxing, conversion to a mixture of
solid-state CuIn$_{1-x}$Ga$_x$Se$_2$, Cu$_2$Se, and uncharacterized In- and
Ga-containing species takes place. Formation of the final nano-
crystalline CuIn$_{1-x}$Ga$_x$Se$_2$ ($0 \leq x \leq 1$) products is complete after
annealing at 500 °C.

SEM images of post-annealed CIGS-1 and CGS products
are shown in Figure 1. These and the CIS, CIGS-2, and CIGS-
3 samples show similar morphologies consisting of mixtures of
plate-like particles or large nodules (100–400 nm in diameter),
nanorods (50–100 nm diameter), and clusters of spherical
nanoparticles in the diameter range of 20–40 nm each. Under
the annealing conditions studied, no thin-film formation was
observed.

XRD data (Figure 2) are consistent with standard values re-
ported for CuIn$_{1-x}$Ga$_x$Se$_2$ crystalline materials [18]. Each product
shows seven orientations in expected 2θ positions for the (1 1 2),
(2 0 4/2 2 0), (1 1 6/3 1 2), (4 0 0), (3 1 6/3 3 2), (4 4 2/2 2 8), and
(5 1 2) crystal planes. The CIGS-3 (2 0 4/2 2 0) and (1 1 6/3 1 2)
peaks exhibit broadening due to the separation of the (2 0 4),
(2 2 0), (1 1 6), and (3 1 2) plane diffractions. In CGS, these four
signals are completely resolved. As gallium concentration in-
creases, all signals shift toward larger diffraction angles. Figure
3 shows this effect on the d(1 1 2) signal. The (1 1 2) lattice plane
spacing correlates linearly with Ga/(Ga+In). Lattice parameters a
and c decrease with increase in gallium concentration, ranging be-
tween those for the CIS ($a = 5.779 \text{ Å}, c = 11.569 \text{ Å}$) and CGS ($a
= 5.617 \text{ Å}, c = 11.051 \text{ Å}$) samples.

All post-annealed products exhibit Raman spectra (Figure 4)
with single, intense scattering peaks between 172 and 185 cm$^{-1}$
corresponding to the A$_1$ optical phonon mode that is charac-
teristic of the chalcopyrite crystal structure [19]. With increase in
Ga content, the peak position shifts to higher frequency and cor-
relates linearly to Ga/(Ga+In) as has been reported for thin-film
CuIn$_{1-x}$Ga$_x$Se$_2$ [20]. In the CIGS-3 and CGS Raman spectra, in-
tense broad peaks are also observed in the 260–275 cm$^{-1}$ range.
Based on literature reports [21] and matching with authentic

![Figure 1](image-url) SEM images of post-annealed (a) CuIn$_{1-x}$Ga$_x$Se$_2$ (CIGS-1 sample) and (b) CuGaSe$_2$ (CGS sample). The white bar lengths equal (a) 300 and (b) 1000 nm.

![Figure 2](image-url) XRD patterns for CIS, CIGS-1, CIGS-2, CIGS-3, and CGS nanocrystalline samples. Intensity signals are offset for clarity.
samples, this peak has been assigned to one or more Cu$_{x}$Se$_{2-2x}$-phase impurities. Close examination of the XRD patterns of these products reveal small impurity signals at 2θ = 44.9° consistent with Cu$_{x}$Se$_{2-2x}$ phases. Depth-profile Auger analyses indicate that the impurities reside on the particle surfaces. In the CGS sample, a separate Raman peak at 274 cm$^{-1}$ corresponding to the CuGaSe$_{2}$ B$_2$ phonon [22] can be discerned.

Employing only the solvothermal reaction step can yield nanocrystalline CuInSe$_2$ of reasonable quality but in CuIn$_{x}$Ga$_{1-x}$Se$_2$ ($x$>0) preparation, there is often significant Cu$_{x}$Se$_{2-2x}$ contamination. The latter annealing step in our two-step process serves to convert this to the final CuIn$_{x}$Ga$_{1-x}$Se$_2$ product. To determine the effects of annealing temperature on this process, the solid CuGaSe$_2$/Cu$_{x}$Se$_{2-2x}$ mixture from a CGS solvothermal preparation reaction was isolated. Portions of this sample were annealed at 200, 300, 400, and 500 °C for 20 min each. Raman spectra (Figure 5) reveal that the portion of Cu$_{x}$Se$_{2-2x}$ converted to CuGaSe$_2$ increases with temperature, with complete conversion occurring at 500 °C or higher. The Raman intensities at 185 cm$^{-1}$ (CuGaSe$_2$, A$_1$ phonon) and 274 cm$^{-1}$ (CuGaSe$_2$, B$_2$ phonon) [22] slightly increase with increase in annealing temperature while the 263 cm$^{-1}$ peak connected with several phases of Cu$_{x}$Se$_{2-2x}$ rapidly disappears. This effect was also observed for the CIS, CIGS-1, CIGS-2, and CIGS-3 samples, but the B$_2$ phonons are much less prominent due to less tetragonal distortion in the crystal lattices [22].

If the Cu$_{x}$Se$_{2-2x}$-to-CuIn$_{1-x}$Ga$_x$Se$_2$ conversion mechanisms are consistent with what has been previously reported for solid-state CuIn$_{1-x}$Ga$_x$Se$_2$ formation [23], direct reactions of Cu$_{x}$Se$_{2-2x}$+In$_x$Se$_3$ (in CuInSe$_2$ formation), CuInSe$_x$+Ga$_{1-x}$Se$_3$ (in CuIn$_{1-x}$Ga$_x$Se$_2$ formation), and Cu$_{x}$Se$_{2-2x}$+Ga$_{1-x}$Se$_3$ (in CuGaSe$_2$ formation) may be responsible. While no In$_x$Se$_3$ or Ga$_x$Se$_3$ species were observed by XRD or Raman in the pre-annealed samples, it must be concluded that in addition to Cu$_{x}$Se$_{2-2x}$ one or more In- and/or Ga-containing species are present. At this point, the nature of these species is unknown. They must be amorphous, as no RDX signals from materials other than CuIn$_{x}$Se$_2$, CuIn$_{1-x}$Ga$_x$Se$_2$, CuGaSe$_2$, and Cu$_{x}$Se$_{2-2x}$ were observed. Although Raman-active phonon frequencies for amorphous In$_{1-x}$Se alloys [24] and GaSe [25] have been reported, none of these were exhibited by our samples. Studies of the solvothermal reaction pathways and the nature of the pre-annealed material products are in progress.

4. Conclusion

We have developed a two-step, non-vacuum process for the preparation of CuIn$_{1-x}$Ga$_x$Se$_2$ nanocrystalline materials of varying In/Ga composition ratios (0 ≤ $x$ ≤ 1). An open-air solvothermal reaction of CuCl$_x$, InCl$_x$ (for $x$ < 1), GaCl$_x$ (for $x$ > 0), and Se in refluxing C$_2$H$_4$N$_2$ was followed by the annealing of the isolated solid product in a nitrogen atmosphere at temperatures between 200 and 500 °C. To the best of our knowledge, this is the first reported open-air solvothermal procedure employed in the preparation of CuIn$_{1-x}$Ga$_x$Se$_2$ ($x$>0) materials. The high boiling point (267 °C) of C$_2$H$_4$N$_2$ appears to facilitate gallium incorporation into the nanocrystalline product structures. Annealing at 500 °C serves to convert precursor solids to the chalcopyrite product and improve the crystallinity of any pre-existing chalcopyrite.
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References


