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Scanning tunneling microscopy study of intermediates in the dissociative adsorption of closo-1,2-dicarbadosicarbaborane on Si(111)

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Closo-1,2-dicarbadodecaborane (C₆B₁₀H₁₂) is a source compound found to be suitable for the deposition of a high resistivity form of boron-carbide (B₄C), and the fabrication of boron-rich semiconductor devices. A scanning tunneling microscope (STM) was used to image these molecular icosahedra on Si(111)-(7×7). Molecular decomposition (tip induced and otherwise) produced a boron-carbide/silicon interface with pronounced heterojunction electronic characteristics. In STM, this interface is characterized by a disordering of the Si(111)-(7×7) reconstruction. We suggest, based on Auger electron spectroscopy data and low-energy electron diffraction observations, that boron atoms from the dissociated source molecules substitutionally occupy selvedge sites, as in the boron-induced (3×3)R30º reconstruction of Si(111). © 1995 American Vacuum Society.

I. INTRODUCTION

The combination of a chemical vapor deposition (CVD) source molecule and a scanning tunneling microscope (STM) has long been considered a promising route to the “direct writing” of nanostructures with varying compositions. In this process, an STM tip programmed to trace an arbitrary pattern performs like a subnanometer-resolution electron gun. The energy-tunable electron flux from this highly positionable source is used to decompose portions of a deposited layer of CVD source molecules. Further processing can afterward be applied to complete the nanofabrication process. Annealing, for example, could be employed to remove any unwanted species/undecomposed molecules through thermal desorption. The final product is a designed array of local regions with altered composition. Both organometallic molecules¹-⁴ and main group carboranes⁵ have thus far been explored as potential CVD source compounds for direct writing in an STM.

For the STM-induced dissociation to be truly a selective area deposition process, the source must adsorb molecularly, rather than dissociatively.¹ In this article, we show that such a molecular precursor state exists over a limited coverage range at room temperature for the icosahedral cage molecule closo-1,2-dicarbadodecarborane (C₆B₁₀H₁₂; denoted as orthocarborane in this paper). In much the same way as has been attempted for nido-decarborane⁶ and NH₃,⁷ we have studied the surface reactions of the Si(111)-(7×7) reconstruction with orthocarborane using an STM. Spectra from the B₄C/Si(111) interface showing anisotropic rectifying behavior will also be presented.

II. EXPERIMENT

Orthocarborane is a commercially available molecule with its two carbon atoms next to each other and only exopolyhedral hydrogen atoms⁸ (see Fig. 1). The orthocarborane was distilled and admitted to the ultrahigh vacuum (UHV) chamber following a procedure described elsewhere.⁹ The final purity of the orthocarborane (Aldrich) was determined by infrared, mass spectral, and nuclear magnetic resonance measurements (purity >98%) and compared with literature values;¹⁰ no isomers were found to be present.

Scanning tunneling microscopy experiments were performed with the JEOL JSTM-4500VT microscopy. Normal imaging and spectroscopy set point values were V_bias = +2 V and I_tunnel = 0.5 nA. Images were acquired in both the traditional “constant current” mode, as well as the “constant height” mode, where the tunneling current feedback loop responds slowly, maintaining only the average tunneling current. STM data shown were processed to remove acoustically and electronically induced noise. Low-energy electron diffraction (LEED) observations were made using conventional rear-view optics. Auger electron spectroscopy (AES) measurements were performed in a different UHV chamber using a double-pass cylindrical mirror analyzer. The substrates used in these experiments were Si(111) polished n-type wafers cut to appropriate geometries. These samples were cleaned in situ by heating to approximately 1150 °C for several seconds, while ensuring that the pressure did not rise above 1.0×10⁻⁹ Torr.

III. RESULTS AND DISCUSSION

After a moderate (1–50 Langmuir; 1 L = 1×10⁻⁶ Torr s), uncalibrated dose of orthocarborane, STM imaged a series of 10–15 Å diameter protrusions atop the 7×7 reconstruction of Si(111) as seen in Fig. 2. The average size of the observed species appears larger than the orthocarborane molecule (5.6 Å). However, this can be explained by considering that in STM images, the apparent lateral diameter of a tall protrusion is a convolution of tip shape/radius and real protrusion size. Also, it is possible that molecular clustering and some...
fragmentation result in the variety of sizes found during imaging. Although molecular orthocarborane is believed to be only weakly bound to metal surfaces, the loss of only a few exopolyhedral hydrogens might substantially increase its heat of adsorption and decrease its mobility. This type of selective ligand loss has been observed with smaller carborane cage molecules. Decaborane decomposition studies suggest that hydrogen is removed as $H_2$ in the adsorption process. The molecular icosahedral species observed is clearly a possible candidate as a precursor state to dissociative adsorption.

Concomitant with continuing orthocarborane adsorption, however, a disordering of the Si$_{111}$-(7×7) surface, brought about through decomposed source molecules, becomes evident as seen in Fig. 3. While tip induced decomposition was also observed, spontaneous decomposition happened readily over time and made selective area deposition of boron carbide extremely difficult to achieve. The surface reconstruction, as recorded by STM, closely resembles the reconstruction of Si$_{111}$ upon exposure to decaborane. A number of studies have investigated the binding site for initial boron coverages. It is now generally accepted for initial decaborane exposures, as with other methods for forming a surface boron layer, the initial coverage of boron (1/3 monolayer) occupies the subsurface site with a coordination of 5 ($S_5$), below a silicon surface adatom. A simple AES study suggests a similar mechanism for our system. Figure 4 shows the AES data for an 80 Langmuir dose of orthocarborane ($C_2B_{10}H_{12}$) on clean Si$_{111}$-(7×7). The upper inset shows similar data for a film of $B_5C$. The relative intensity of the boron Auger electron feature to the carbon Auger electron feature differs substantially for these two plots, indicating the loss of some boron by either desorption or absorption. This later explanation is consistent with the surface disordering seen with STM, and our LEED observations. After rapidly heating an orthocarborane-exposed sample to approximately 800 °C, weak but definite $(\sqrt{3} \times \sqrt{3})R30^\circ$ LEED spots became visible. However, unlike the $(\sqrt{3} \times \sqrt{3})R30^\circ$ produced with decaborane, STM observations of the surface show only clumpy disorder, most likely the result of silicon carbide formed with the two cage carbon atoms during annealing. Decaborane contains no carbon, and therefore produces a pure, well-ordered $(\sqrt{3} \times \sqrt{3})R30^\circ$.

Local spectroscopy done with an STM gives verification of the heterojunction behavior of this boron carbide/silicon interface. Figure 5 shows three $I–V$ curves. One is from a clean 7×7 reconstruction and shows this surface’s metallic character. In contrast, data from disordered regions produced
Langmuir's orthocarborane. The bottom insert has been expanded to better
show the boron and carbon lines. The top insert is a reference spectrum of
B₃C (Ref. 8).

The goal now is to produce localized regions with such
altered electronic properties. Although orthocarborane exists
initially in the molecular form on Si(111), and tip-induced
decomposition was achieved, at room temperature decompo-
sition unavoidably occurred over time. Therefore, further re-
finement of the "direct writing" procedure, including the use
of a cold substrate for initial exposure and tunneling
electron-induced decomposition, will be required to produce
localized "boron-carbide diodes" surrounded by clean met-
3 allic 7×7. The results of our room-temperature efforts sug-
uggest that length scales on the order of 10 nm or less will be
readily achievable.

IV. CONCLUSIONS

A complex interface of a boron delta-doped silicon layer
and a carbon-rich boron-carbide layer is formed at the
B₃C/Si(111) heterojunction interface. Avoiding formation of
this complex interface will require passivation of the Si(111)-(7×7)
reconstruction. A consequence of this complex inter-
face is that a B₃C/Si(111) heterojunction may form a p-i-n
junction, instead of a p-n junction. We believe that the boron
occupies a subsurface site within the silicon.

Largely complete orthocarborane icosahedral molecules
may be imaged on Si(111)-(7×7), This may be the precursor
state to dissociative adsorption. The possible loss of one or
two hydrogens from such species cannot, however, be ex-
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Fig. 4. An Auger electron spectrum of Si(111) following exposure to 80
Langmuirs orthocarborane. The bottom insert has been expanded to better
show the boron and carbon lines. The top insert is a reference spectrum of
B₃C (Ref. 8).

Fig. 5. I–V spectra (dotted and dashed), acquired atop disordered regions
where tip-induced orthocarborane decomposition has occurred, clearly show
various degrees of heterojunction characteristics. Included for reference are
data (solid) from the clean metallic Si(111)-(7×7) surface.

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