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Scanning tunneling microscopy study of intermediates in the dissociative adsorption of closo-1,2-dicarbadoodecaborane on Si(111)

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Closo-1,2-dicarbadoodecaborane ($C_2B_{10}H_{12}$) is a source compound found to be suitable for the deposition of a high resistivity form of boron-carbide (B_5C), and the fabrication of boron-rich semiconductor devices. A scanning tunneling microscope (STM) was used to image these molecular icosahedra on Si(111)-(7 \times 7). Molecular decomposition (tip induced and otherwise) produced a boron-carbide/silicon interface with pronounced heterojunction electronic characteristics. In STM, this interface is characterized by a disordering of the Si(111)-(7 \times 7) reconstruction. We suggest, based on Auger electron spectroscopy data and low-energy electron diffraction observations, that boron atoms from the dissociated source molecules substitutionally occupy selvedge sites, as in the boron-induced ($\sqrt{3}\times\sqrt{3}$)R30 $^\circ$ reconstruction of Si(111). © 1995 American Vacuum Society.

I. INTRODUCTION

The combination of a chemical vapor deposition (CVD) source molecule and a scanning tunneling microscope (STM) has long been considered a promising route to the "direct writing" of nanostructures with varying compositions. In this process, an STM tip programmed to trace an arbitrary pattern performs like a subnanometer-resolution electron gun. The energy-tunable electron flux from this highly positionable source is used to decompose portions of a deposited layer of CVD source molecules. Further processing can afterward be applied to complete the nanofabrication process. Annealing, for example, could be employed to remove any unwanted species/undecomposed molecules through thermal desorption. The final product is a designed array of local regions with altered composition. Both organometallic compounds¹⁻⁴ and main group carboranes⁵ have thus far been explored as potential CVD source compounds for direct writing in an STM.

For the STM-induced dissociation to be truly a selective area deposition process, the source must adsorb molecularly, rather than dissociatively.⁵ In this article, we show that such a molecular precursor state exists over a limited coverage range at room temperature for the icosahedral cage molecule closo-1,2-dicarbadoodecaborane ($C_2B_{10}H_{12}$; denoted as orthocarborane in this paper). In much the same way as has been attempted for nido-decaborane⁶ and NH_3 ,⁷ we have studied the surface reactions of the Si(111)-(7 \times 7) reconstruction with orthocarborane using an STM. Spectra from the B_5C /Si(111) interface showing anisotropic rectifying behavior will also be presented.

II. EXPERIMENT

Orthocarborane is a commercially available molecule with its two carbon atoms next to each other and only exopolyhedral hydrogen atoms⁸ (see Fig. 1). The orthocarborane

was distilled and admitted to the ultrahigh vacuum (UHV) chamber following a procedure described elsewhere.⁹ The final purity of the orthocarborane (Aldrich) was determined by infrared, mass spectral, and nuclear magnetic resonance measurements (purity >98%) and compared with literature values;¹⁰ no isomers were found to be present.

Scanning tunneling microscopy experiments were performed with the JEOL JSTM-4500VT microscopy. Normal imaging and spectroscopy set point values were $V_{bias}=+2$ V and $I_{tunnel}=0.5$ nA. Images were acquired in both the traditional "constant current" mode, as well as the "constant height" mode, where the tunneling current feedback loop responds slowly, maintaining only the average tunneling current. STM data shown were processed to remove acoustically and electronically induced noise. Low-energy electron diffraction (LEED) observations were made using conventional rear-view optics. Auger electron spectroscopy (AES) measurements were performed in a different UHV chamber using a double-pass cylindrical mirror analyzer. The substrates used in these experiments were Si(111) polished *n*-type wafers cut to appropriate geometries. These samples were cleaned *in situ* by heating to approximately 1150 $^\circ$ C for several seconds, while ensuring that the pressure did not rise above 1.0×10^{-9} Torr.

III. RESULTS AND DISCUSSION

After a moderate (1–50 Langmuir; $1\text{ L}=1\times 10^{-6}$ Torr s), uncalibrated dose of orthocarborane, STM imaged a series of 10–15 Å diameter protrusions atop the 7 \times 7 reconstruction of Si(111) as seen in Fig. 2. The average size of the observed species appears larger than the orthocarborane molecule (5.6 Å). However, this can be explained by considering that in STM images, the apparent lateral diameter of a tall protrusion is a convolution of tip shape/radius and real protrusion size. Also, it is possible that molecular clustering and some

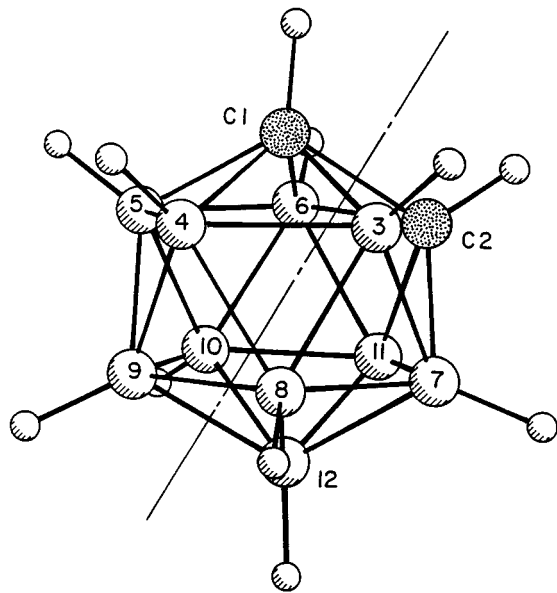


FIG. 1. Schematic diagram of *closo*-1,2-dicarbododecaborane (orthocarborane), with a calculated radius of 5.6 Å.

fragmentation result in the variety of sizes found during imaging. Although molecular orthocarborane is believed to be only weakly bound to metal surfaces,⁹ the loss of only a few exopolyhedral hydrogens might substantially increase its heat of adsorption and decrease its mobility. This type of selective ligand loss has been observed with smaller carborane cage molecules.¹¹ Decaborane decomposition studies¹² suggest that hydrogen is removed as H₂ in the adsorption process. The molecular icosahedral species observed is clearly a possible candidate as a precursor state to dissociative adsorption.

Concomitant with continuing orthocarborane adsorption, however, a disordering of the Si(111)-(7×7) surface, brought about through decomposed source molecules, becomes evi-

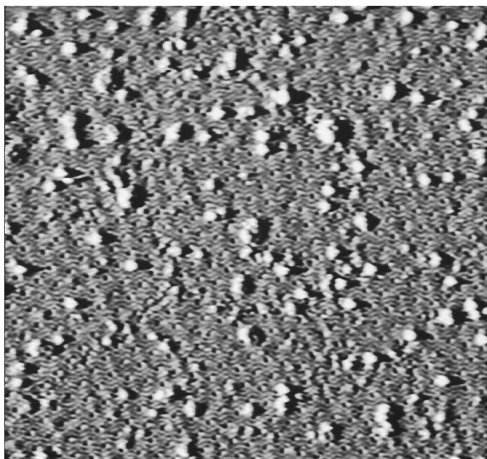


FIG. 2. 800×730 Å² constant-height STM image of orthocarborane species atop the Si(111)-(7×7) reconstruction.

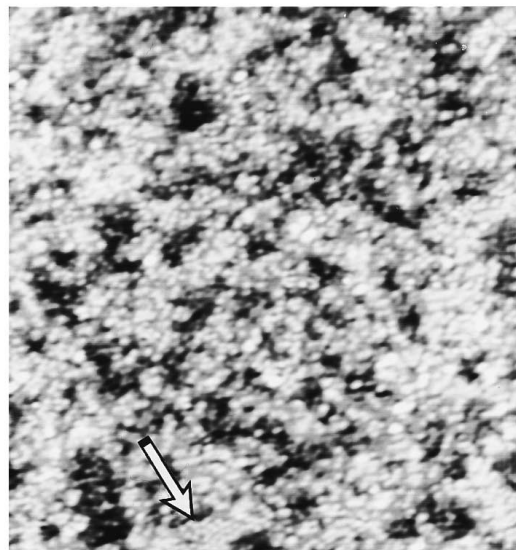


FIG. 3. 470×500 Å² constant-current STM image of the disordered Si(111) surface following long exposure to orthocarborane. An arrow marks a region where remnants of the 7×7 unit cell can still be identified. Full height contrast in this image is 4.7 Å.

dent as seen in Fig. 3. While tip induced decomposition was also observed, spontaneous decomposition happened readily over time and made selective area deposition of boron carbide extremely difficult to achieve. The surface reconstruction, as recorded by STM, closely resembles the reconstruction of Si(111) upon exposure to decaborane.⁶ A number of studies have investigated the binding site for initial boron coverages.^{6,12–20} It is now generally accepted for initial decaborane exposures,^{6,12,17,20} as with other methods for forming a surface boron layer, the initial coverage of boron (1/3 monolayer) occupies the subsurface site with a coordination of 5 (S₅), below a silicon surface adatom.^{6,13–17} A simple AES study suggests a similar mechanism for our system. Figure 4 shows the AES data for an 80 Langmuir dose of orthocarborane (C₂B₁₀H₁₂) on clean Si(111)-(7×7). The upper inset shows similar data for a film of B₅C. The relative intensity of the boron Auger electron feature to the carbon Auger electron feature differs substantially for these two plots, indicating the loss of some boron by either desorption or absorption. This later explanation is consistent with the surface disordering seen with STM, and our LEED observations. After rapidly heating an orthocarborane-exposed sample to approximately 800 °C, weak but definite ($\sqrt{3}\times\sqrt{3}$)R30° LEED spots became visible. However, unlike the ($\sqrt{3}\times\sqrt{3}$)R30° produced with decaborane, STM observations of the surface show only clumpy disorder, most likely the result of silicon carbide formed with the two cage carbon atoms during annealing. Decaborane contains no carbon, and therefore produces a pure, well-ordered ($\sqrt{3}\times\sqrt{3}$)R30°.

Local spectroscopy done with an STM gives verification of the heterojunction behavior of this boron carbide/silicon interface. Figure 5 shows three *I*–*V* curves. One is from a clean 7×7 reconstruction and shows this surface's metallic character. In contrast, data from disordered regions produced

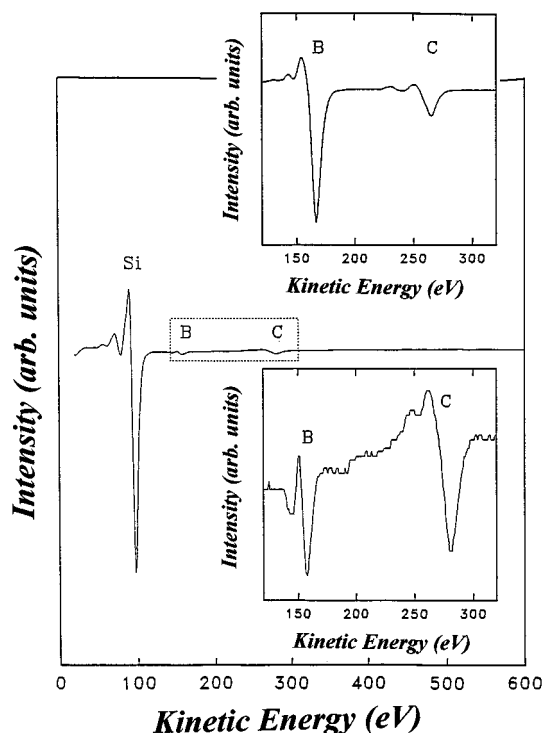


FIG. 4. An Auger electron spectrum of Si(111) following exposure to 80 Langmuirs orthocarborane. The bottom insert has been expanded to better show the boron and carbon lines. The top insert is a reference spectrum of B_5C (Ref. 8).

upon orthocarborane decomposition show a range of anisotropy. One such $I-V$ trace is very similar to boron carbide/ n -Si(111) diodes formed by more conventional methods.²¹

The goal now is to produce localized regions with such altered electronic properties. Although orthocarborane exists initially in the molecular form on Si(111), and tip-induced decomposition was achieved, at room temperature decomposition unavoidably occurred over time. Therefore, further refinement of the “direct writing” procedure, including the use

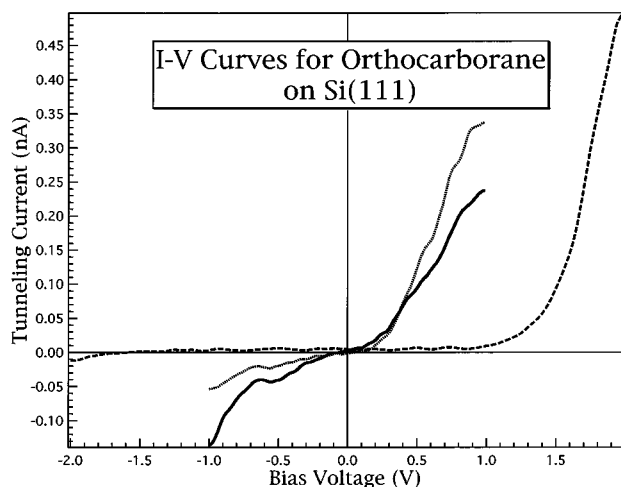


FIG. 5. $I-V$ spectra (dotted and dashed), acquired atop disordered regions where tip-induced orthocarborane decomposition has occurred, clearly show various degrees of heterojunction characteristics. Included for reference are data (solid) from the clean metallic Si(111)-(7 \times 7) surface.

of a cold substrate for initial exposure and tunneling electron-induced decomposition, will be required to produce localized “boron-carbide diodes” surrounded by clean metallic 7 \times 7. The results of our room-temperature efforts suggest that length scales on the order of 10 nm or less will be readily achievable.

IV. CONCLUSIONS

A complex interface of a boron delta-doped silicon layer and a carbon-rich boron-carbide layer is formed at the $B_5C/Si(111)$ heterojunction interface. Avoiding formation of this complex interface will require passivation of the Si(111)-(7 \times 7) reconstruction. A consequence of this complex interface is that a $B_5C/Si(111)$ heterojunction may form a $p-i-n$ junction, instead of a $p-n$ junction. We believe that the boron occupies a subsurface site within the silicon.

Largely complete orthocarborane icosahedral molecules can be imaged on Si(111)-(7 \times 7). This may be the precursor state to dissociative adsorption. The possible loss of one or two hydrogens from such species cannot, however, be excluded.

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¹Cd(Me)₂: E. E. Ehrichs, S. Yoon, and A. L. de Lozanne, Appl. Phys. Lett. **53**, 2287 (1988); E. E. Ehrichs, R. M. Silver, and A. L. de Lozanne, J. Vac. Sci. Technol. A **6**, 540 (1988); R. M. Silver, E. E. Ehrichs, and A. L. de Lozanne, Appl. Phys. Lett. **51**, 247 (1987).

²Fe(CO)₅: A. D. Kent, T. M. Shaw, S. von Molnár, and D. D. Awschalom, Science **262**, 1249 (1993); M. A. McCord and D. D. Awschalom, Appl. Phys. Lett. **57**, 2153 (1990); D. D. Awschalom, M. A. McCord, and G. Grinstein, Phys. Rev. Lett. **65**, 783 (1990).

³Al(Me)₃: E. E. Ehrich and A. L. de Lozanne, in *Nanostructure Physics and Fabrication*, edited by M. A. Reed and W. P. Kirk (Academic, New York, 1989), p. 441.

⁴Pd(η^3 -C₃H₅)(η^5 -C₅H₅): D. S. Saulys, A. Ermakov, E. L. Garfunkel, and P. A. Dowben, J. Appl. Phys. **76**, 7639 (1994).

⁵B₁₀H₁₄: G. Dujardin, R. E. Walkup, and Ph. Avouris, Science **255**, 1232 (1992); F. K. Perkins, M. Onellion, S. Lee, and P. A. Dowben, *Photons and Low-Energy Particles in Surface Processing*, edited by C. I. H. Ashby, J. H. Brannon, and S. W. Pang, Mater. Res. Soc. Symp. Proc. **236**, 153 (1992).

⁶Ph. Avouris, J. Phys. Chem. **94**, 2246 (1990); Ph. Avouris, In-Whan Lyo, F. Bozso, and E. Kaxiras, J. Vac. Sci. Technol. A **8**, 3405 (1990); In-Whan Lyo, E. Kaxiras, and Ph. Avouris, Phys. Rev. Lett. **63**, 1261 (1989).

⁷Ph. Avouris and R. Wolkow, Phys. Rev. B **39**, 5091 (1989).

⁸F. Lee, P. A. Dowben, A. T. Wen, A. P. Hitchcock, J. A. Glass, and J. T. Spencer, J. Vac. Sci. Technol. A **10**, 881 (1992).

⁹D. Byun, S.-D. Hwang, P. A. Dowben, F. K. Perkins, F. Filips, and N. J. Ianno, Appl. Phys. Lett. **64**, 1968 (1994); H. Zeng, D. Byun, J. Zhang, G. Vidali, M. Onellion, and P. A. Dowben, Surf. Sci. **313**, 239 (1994).

¹⁰A. P. Hitchcock, A. T. Wen, S. Lee, J. A. Glass, J. T. Spencer, and P. A. Dowben, J. Phys. Chem. **97**, 8171 (1993), and the references therein.

¹¹S. Lee, D. Li, P. A. Dowben, F. K. Perkins, M. Onellion, and J. T. Spencer, J. Am. Chem. Soc. **113**, 8444 (1991).

¹²P. J. Chen, M. L. Colaianni, and J. T. Yates, J. Appl. Phys. **72**, 3155 (1992).

- ¹³E. Kaxiras, K. C. Pandey, F. J. Himpsel, and R. M. Tromp, *Phys. Rev. B* **41**, 1262 (1990).
- ¹⁴P. Bedrossian, R. D. Mead, K. Mortensen, D. M. Chen, J. A. Golovchenko, and D. Vanderbilt, *Phys. Rev. Lett.* **63**, 1257 (1989).
- ¹⁵R. L. Headrick, I. K. Robinson, E. Vlieg, and L. C. Feldman, *Phys. Rev. Lett.* **63**, 1253 (1989).
- ¹⁶A. B. McLean, L. J. Terminello, and F. J. Himpsel, *Phys. Rev. B* **4**, 7694 (1990); H. Hwang, S. Y. Tong, J. Quinn, and F. Jona, *ibid.* **41**, 3276 (1990).
- ¹⁷H. H. Weitering, J. Chen, N. J. Di Nardo, and E. W. Plummer, *Phys. Rev. B* **48**, 8119 (1993).
- ¹⁸Y. Ma, J. E. Rowe, E. E. Chaban, C. T. Chen, R. L. Headrich, G. M. Meigs, S. Modesti, and F. Sette, *Phys. Rev. Lett.* **65**, 2172 (1990).
- ¹⁹J. E. Rowe, G. K. Wertheim, and D. M. Riffe, *J. Vac. Sci. Technol. A* **9**, 1020 (1991).
- ²⁰R. A. Rosenberg, S. P. Frigo, S. Lee, and P. A. Dowben, *J. Appl. Phys.* **71**, 4795 (1992).
- ²¹S. Lee and P. A. Dowben, *Appl. Phys. A* **58**, 223 (1994).